



# Niobium oxide electrode performance boosted by molybdenum doping and calcination for supercapacitor applications

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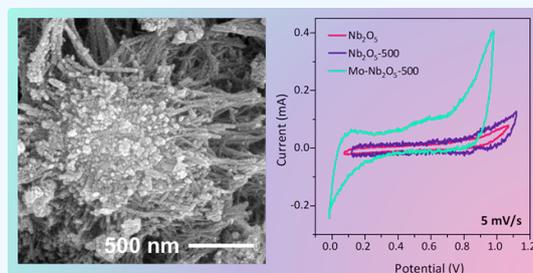
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## ABSTRACT

Niobium pentoxide ( $\text{Nb}_2\text{O}_5$ ) is a promising pseudocapacitive material for supercapacitor applications due to its high theoretical capacitance and electrochemical stability. However, its practical performance is hindered by low electrical conductivity and limited ion transport kinetics. In this study, we report the enhancement of  $\text{Nb}_2\text{O}_5$  electrode performance through molybdenum (Mo) doping and thermal calcination. Mo-doped  $\text{Nb}_2\text{O}_5$  nanostructures were synthesized via a hydrothermal method followed by calcination at 500 °C. Scanning electron microscopy (SEM) and energy-dispersive X-ray spectroscopy (EDS) confirmed the formation of a rougher surface morphology and homogeneous Mo distribution in the doped sample. X-ray diffraction (XRD) analysis revealed a phase transition from a distorted orthorhombic structure in pristine  $\text{Nb}_2\text{O}_5$  to a more crystalline pseudo-hexagonal phase in Mo- $\text{Nb}_2\text{O}_5$ -500. Electrochemical measurements demonstrated significantly enhanced capacitive behavior, with Mo- $\text{Nb}_2\text{O}_5$ -500 achieving a specific capacitance of 55.3  $\text{F g}^{-1}$  at a scan rate of 5  $\text{mV s}^{-1}$ , approximately five times higher than that of the undoped material. All electrodes exhibited stable cycling performance. These results highlight the synergistic effect of Mo doping and calcination in improving the electrochemical properties of  $\text{Nb}_2\text{O}_5$ , providing a viable strategy for the development of high-performance pseudocapacitor electrodes.



**Keywords:** Niobium pentoxide ( $\text{Nb}_2\text{O}_5$ ); Molybdenum doping; Calcination; Pseudocapacitor; Electrochemical performance

## 1. INTRODUCTION

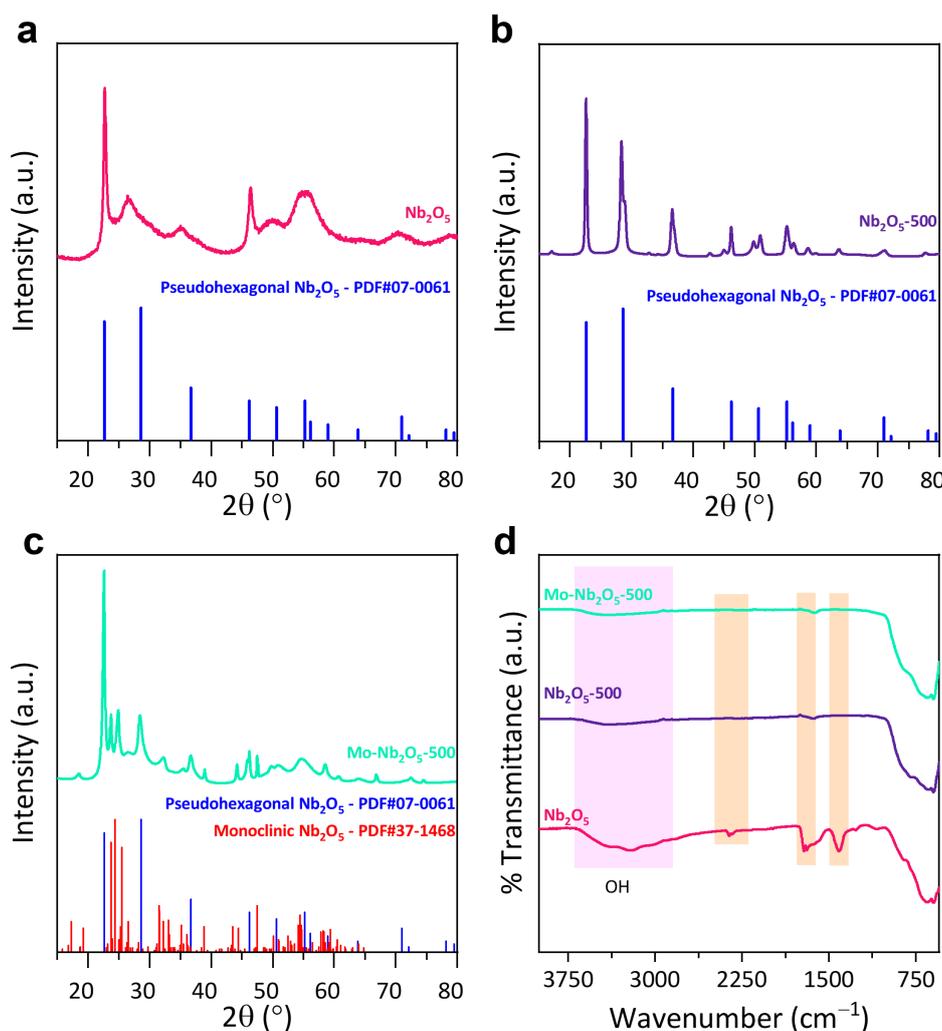
The rapid advancement of energy storage technologies has driven considerable interest in developing high-performance electrode materials for supercapacitors [1, 2]. Among various transition metal oxides, materials such as ruthenium dioxide ( $\text{RuO}_2$ ) [3], manganese dioxide ( $\text{MnO}_2$ ) [4, 5], cobalt oxide ( $\text{Co}_3\text{O}_4$ ) [6, 7], and nickel oxide ( $\text{NiO}$ ) [8, 9] have been extensively investigated for supercapacitor applications owing to their high specific capacitance and pseudocapacitive behavior. However, issues such as high cost (e.g.,  $\text{RuO}_2$ ), poor conductivity (e.g.,  $\text{MnO}_2$ ), or limited cycle life (e.g.,  $\text{NiO}$  and  $\text{Co}_3\text{O}_4$ ) constrain their practical application. In this context, niobium pentoxide ( $\text{Nb}_2\text{O}_5$ ) has emerged as a promising candidate due to its high theoretical capacitance, wide electrochemical window, and excellent structural stability [10]. However, the practical application of  $\text{Nb}_2\text{O}_5$  in supercapacitors remains limited by its inherently low electronic conductivity and relatively poor ionic diffusion kinetics, which result in suboptimal specific capacitance and rate performance [11].

To overcome these limitations, two principal strategies have been extensively explored: heteroatom doping and thermal treatment (calcination). Doping with transition metals such as molybdenum (Mo), tungsten (W), or vanadium (V) has been shown to effectively modulate the electronic struc-

ture of  $\text{Nb}_2\text{O}_5$ , introduce additional redox-active sites, and enhance electrical conductivity [12–14]. In particular, Mo doping can introduce donor states near the conduction band, facilitating faster electron transport and enabling pseudocapacitive charge storage through reversible faradaic reactions [15, 16]. Furthermore, doping may also influence the material's morphology and crystallinity, promoting more accessible surface area and optimized ion diffusion pathways.

Calcination, on the other hand, plays a critical role in improving the crystallinity and structural integrity of the synthesized  $\text{Nb}_2\text{O}_5$ . Controlled thermal treatment can reduce structural defects, enhance grain connectivity, and tune porosity, all of which contribute to improved electrochemical performance [17–19]. When combined with doping, calcination can synergistically reinforce the beneficial effects by stabilizing dopant distribution and promoting favorable microstructural evolution.

Despite previous efforts, the combined effect of Mo doping and calcination on the microstructure, elemental distribution, and electrochemical performance of  $\text{Nb}_2\text{O}_5$  has not been fully elucidated. In this work, we synthesize Mo-doped  $\text{Nb}_2\text{O}_5$  via a hydrothermal method followed by calcination and systematically investigate the impact of both modifications on the material's morphology, composition, and supercapacitor performance. The findings provide valuable



**Figure 1.** (a–c) X-ray diffraction (XRD) patterns of pristine  $\text{Nb}_2\text{O}_5$ , thermally treated  $\text{Nb}_2\text{O}_5$  ( $\text{Nb}_2\text{O}_5$ -500), and Mo-doped  $\text{Nb}_2\text{O}_5$  after calcination ( $\text{Mo-Nb}_2\text{O}_5$ -500), illustrating phase evolution and crystallinity changes; (d) Fourier-transform infrared (FTIR) spectra of the same samples, showing vibrational features and changes in bonding environment induced by calcination and Mo incorporation.

insights into the structure–property relationships in doped Nb-based oxides and offer a viable strategy for designing advanced electrode materials with enhanced capacitive behavior.

## 2. MATERIALS AND METHODS

### 2.1 Materials

High-purity ammonium niobate(V) oxalate hydrate ( $\text{ANO}$ ,  $\text{C}_4\text{H}_4\text{NNbO}_9 \cdot x\text{H}_2\text{O}$ , 99.99% trace metals basis) and ammonium molybdate tetrahydrate ( $\text{AMT}$ ,  $(\text{NH}_4)_6\text{Mo}_7\text{O}_{24} \cdot 4\text{H}_2\text{O}$ , 99.98% trace metals basis) were obtained from Sigma-Aldrich, Singapore. Deionized (DI) water was used as solvent during the hydrothermal process.

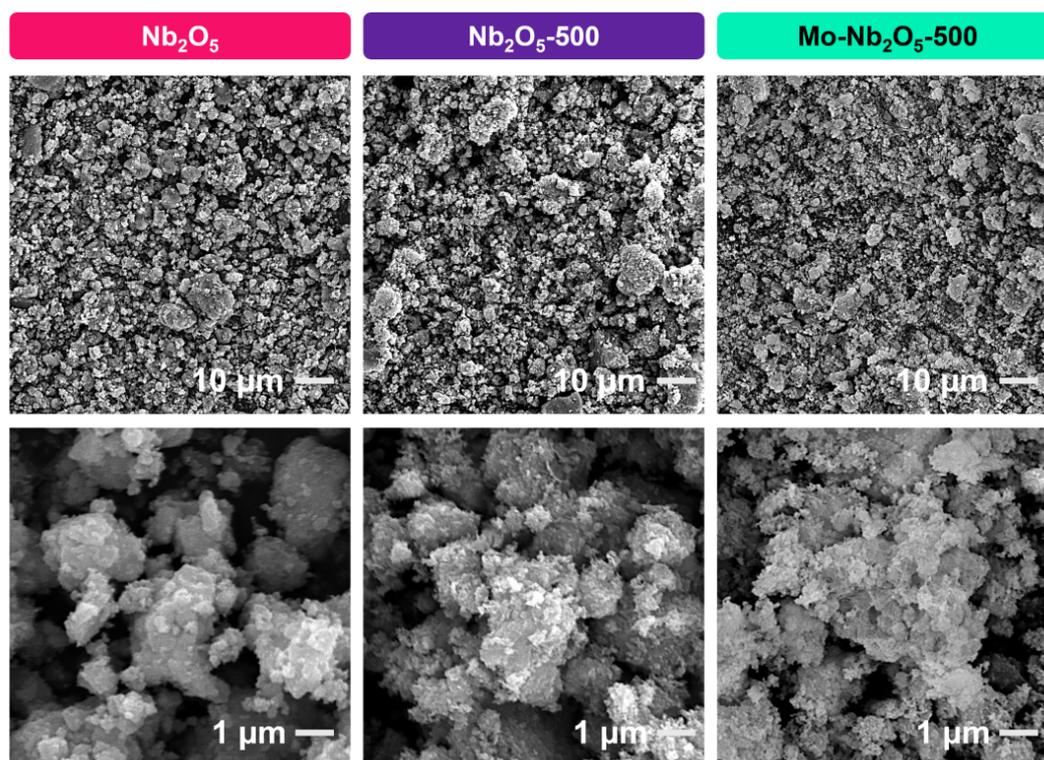
### 2.2 Preparation of $\text{Mo-Nb}_2\text{O}_5$ Nanoparticles

Mo-doped  $\text{Nb}_2\text{O}_5$  nanostructures were synthesized via a hydrothermal method, adapted from previously reported studies [20]. In a typical synthesis, 3.4 g of ammonium niobate(V) oxalate hydrate were dissolved in 70 mL of DI water under continuous magnetic stirring at 600 rpm to ensure complete dissolution. Once a clear and homogeneous solution was

obtained, 0.3 g of ammonium molybdate tetrahydrate was introduced as the molybdenum source. The resulting solution was transferred into a 100-mL Teflon-lined stainless-steel autoclave and subjected to hydrothermal treatment at 175 °C for 72 hours. Upon completion of the reaction, the precipitate was collected by vacuum filtration, thoroughly washed with deionized water to eliminate residual impurities, and dried in an oven at 80 °C for 6 hours. To improve crystallinity and electrochemical properties, the dried product underwent calcination in ambient air at 500 °C for 2 hours. This final sample was designated as  $\text{Mo-Nb}_2\text{O}_5$ -500. For comparative analysis, two control samples were also prepared: undoped  $\text{Nb}_2\text{O}_5$  subjected to identical calcination conditions ( $\text{Nb}_2\text{O}_5$ -500), and uncalcined, undoped  $\text{Nb}_2\text{O}_5$  ( $\text{Nb}_2\text{O}_5$ ).

### 2.3 Materials Characterizations

The morphological and microstructural features of the synthesized samples were examined using scanning electron microscopy (SEM) performed on a JEOL JSM-6510 microscope equipped with an energy-dispersive X-ray spectroscopy (EDS) detector for elemental analysis and mapping. High-resolution



**Figure 2.** Scanning electron microscopy (SEM) images of Nb<sub>2</sub>O<sub>5</sub>, Nb<sub>2</sub>O<sub>5</sub>-500, and Mo-Nb<sub>2</sub>O<sub>5</sub>-500 samples at low (top row, 10 μm) and high (bottom row, 1 μm) magnifications, illustrating morphological evolution induced by thermal calcination and Mo doping. Notable changes include surface coarsening, increased particle definition, and textural roughening in the doped and calcined samples.

surface imaging was further conducted using field-emission scanning electron microscopy (FESEM) on a JEOL JSM-7610F to assess the finer details of particle morphology and surface texture. Phase composition and crystallographic structure were analyzed using X-ray diffraction (XRD) measurements carried out on a Rigaku SmartLab SE Basic diffractometer equipped with a Cu K $\alpha$  radiation source ( $\lambda = 1.5406 \text{ \AA}$ ). The XRD patterns were recorded over an appropriate  $2\theta$  range to identify phase purity and assess the effect of calcination and Mo doping on the crystal structure of the Nb<sub>2</sub>O<sub>5</sub>-based materials.

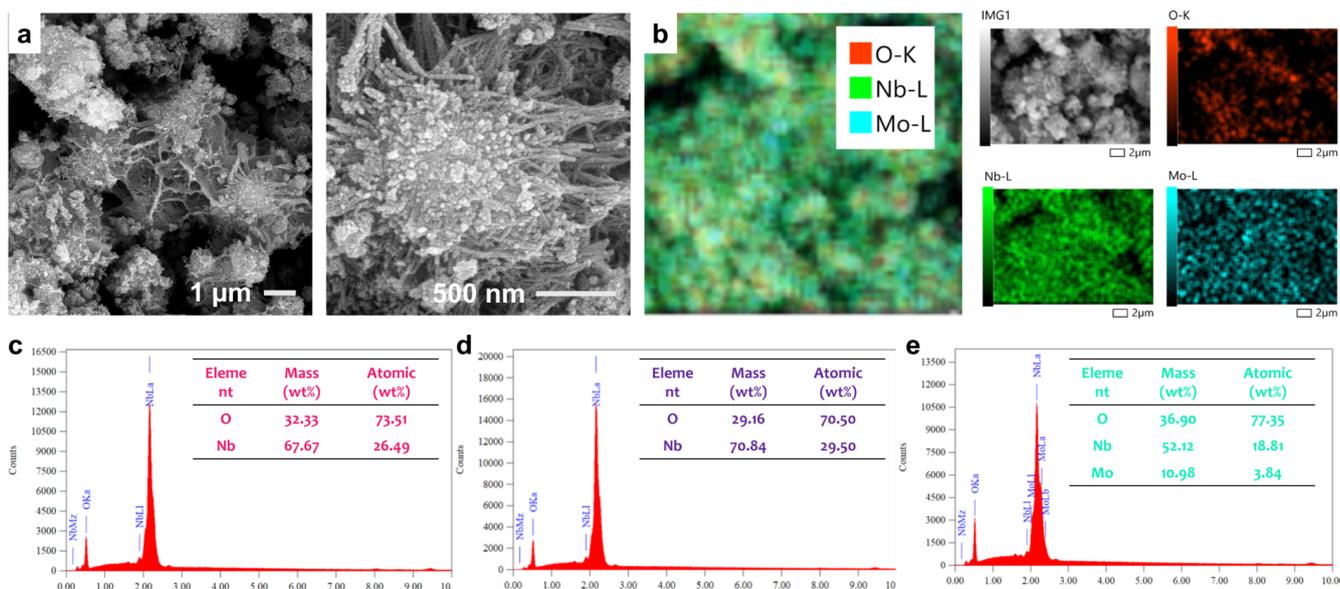
## 2.4 Electrochemical Investigation

Cyclic voltammetry (CV) measurements were conducted to evaluate the electrochemical performance of Nb<sub>2</sub>O<sub>5</sub>, Nb<sub>2</sub>O<sub>5</sub>-500, and Mo-Nb<sub>2</sub>O<sub>5</sub>-500 electrodes. The electrode slurry was prepared by mixing each active material (0.038 g) with PVDF binder (0.0127 g) and acetylene black (0.0127 g), followed by gradual addition of N-methyl-2-pyrrolidone (76 μL total, approximately 10 drops). The mixture was magnetically stirred for 1.5 h to obtain a homogeneous dispersion. Subsequently, 3 μL of slurry was drop-cast onto a glassy carbon electrode (GCE) and dried at 60 °C for 30 minutes to form the working electrode. CV tests were performed in 1 M Na<sub>2</sub>SO<sub>4</sub> electrolyte using a three-electrode configuration: GCE as the working electrode, Ag/AgCl as the reference electrode, and Pt wire as the counter electrode. The measurements were carried out using a XPOW-AX-PST Potentiostat (Nicslab, Indonesia), and the obtained voltammograms were used to derive the effective specific capacitance of each sample [21, 22].

## 3. RESULTS AND DISCUSSION

The crystalline structures of Nb<sub>2</sub>O<sub>5</sub>, Nb<sub>2</sub>O<sub>5</sub>-500, and Mo-Nb<sub>2</sub>O<sub>5</sub>-500 were analyzed using X-ray diffraction (XRD), with the results presented in Figure 1a–c. The uncalcined Nb<sub>2</sub>O<sub>5</sub> sample (Figure 1a) displays broad and relatively low-intensity peaks that resemble a deformed orthorhombic structure, indicative of poor crystallinity and potential structural disorder typically associated with as-synthesized Nb<sub>2</sub>O<sub>5</sub> from hydrothermal methods [23, 24]. Such features suggest partial amorphization or the presence of nanocrystalline domains with local lattice distortions. Upon calcination, the XRD pattern of the Nb<sub>2</sub>O<sub>5</sub>-500 sample (Figure 1b) reveals significantly sharpened and more intense peaks that match well with the pseudohexagonal Nb<sub>2</sub>O<sub>5</sub> phase (PDF#07-0061) [25, 26]. This suggests a substantial improvement in crystallinity, driven by thermal energy promoting atomic rearrangement and long-range structural ordering. The emergence of well-resolved peaks confirms that the calcination process effectively transforms the initially disordered structure into a more ordered crystalline phase.

Further enhancement is observed in the Mo-doped and calcined sample (Figure 1c). Mo-Nb<sub>2</sub>O<sub>5</sub>-500 displays even sharper peaks, consistent with the pseudohexagonal Nb<sub>2</sub>O<sub>5</sub> phase and no secondary phases such as MoO<sub>3</sub>, indicating successful Mo incorporation into the Nb<sub>2</sub>O<sub>5</sub> lattice without inducing phase segregation. Additionally, minor features corresponding to the monoclinic phase (PDF#37-1468) are discernible [27, 28], suggesting a possible coexistence or intermediate phase during lattice reorganization. The preserved pseudohexagonal structure and improved crystallinity are



**Figure 3.** (a) Field emission scanning electron microscopy (FESEM) image showing the microstructure of Mo-Nb<sub>2</sub>O<sub>5</sub>-500; (b) energy-dispersive X-ray spectroscopy (EDS) elemental mapping confirming the uniform spatial distribution of oxygen (O), niobium (Nb), and molybdenum (Mo); (c-e) EDS spectra and corresponding elemental compositions for Nb<sub>2</sub>O<sub>5</sub>, Nb<sub>2</sub>O<sub>5</sub>-500, and Mo-Nb<sub>2</sub>O<sub>5</sub>-500, respectively, illustrating compositional evolution through calcination and Mo doping.

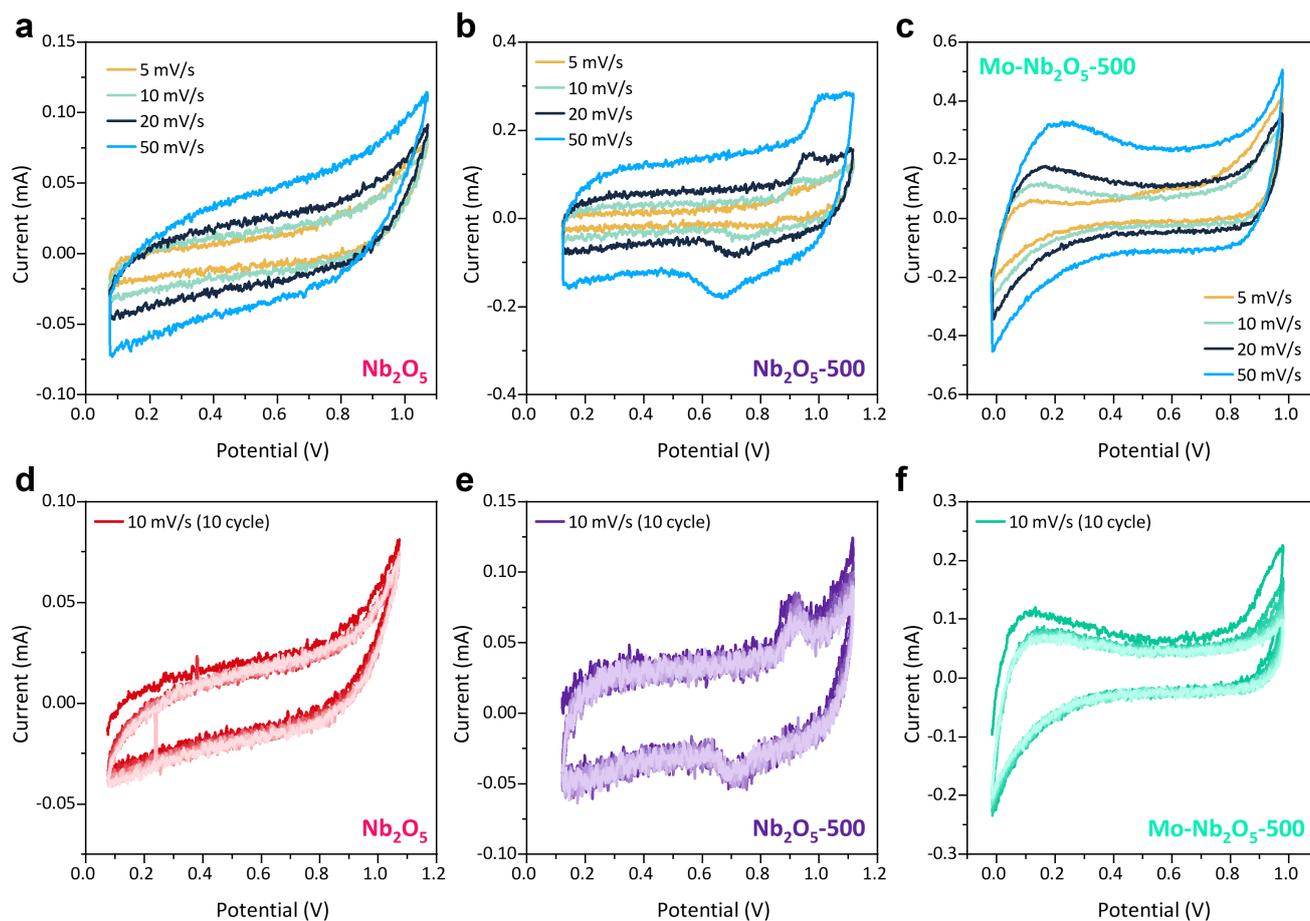
likely facilitated by both Mo doping and calcination. The close ionic radii of Mo<sup>6+</sup> and Nb<sup>5+</sup> support lattice substitution without significant distortion [16]. Previous studies have demonstrated that the incorporation of Mo atoms into the Nb<sub>2</sub>O<sub>5</sub> lattice induces changes in lattice parameters, primarily attributed to an increase in interplanar (*d*) spacing [16]. Overall, the transition from a broad, amorphous-like pattern in Nb<sub>2</sub>O<sub>5</sub> to well-defined crystalline features in Mo-Nb<sub>2</sub>O<sub>5</sub>-500 confirms the critical roles of both calcination and doping. The structural evolution toward a highly ordered pseudohexagonal phase is expected to enhance electron mobility, ion diffusion, and ultimately, electrochemical performance.

The Fourier-transform infrared (FTIR) spectra of Nb<sub>2</sub>O<sub>5</sub>, Nb<sub>2</sub>O<sub>5</sub>-500, and Mo-Nb<sub>2</sub>O<sub>5</sub>-500 (Figure 1d) are dominated by bands in the 900–500 cm<sup>-1</sup> region, which are assigned to Nb–O stretching and Nb–O–Nb bridging vibrations of NbO<sub>6</sub> octahedra, characteristic of Nb<sub>2</sub>O<sub>5</sub>-based oxides. The overall shape and positions of these Nb–O-related bands are very similar for all three samples, indicating that neither calcination nor Mo doping alters the primary Nb<sub>2</sub>O<sub>5</sub> framework. This is consistent with previous Mo-doped Nb<sub>2</sub>O<sub>5</sub> work, where doping did not introduce new vibrational features but preserved the Nb<sub>2</sub>O<sub>5</sub> lattice [16]. A broad absorption band between 3750 and 3000 cm<sup>-1</sup> is observed for the as-prepared Nb<sub>2</sub>O<sub>5</sub>, associated with OH stretching of surface hydroxyl groups and adsorbed water, accompanied by a weaker band near 1630 cm<sup>-1</sup> from H–O–H bending. Both features are markedly reduced in Nb<sub>2</sub>O<sub>5</sub>-500 and Mo-Nb<sub>2</sub>O<sub>5</sub>-500, confirming that calcination at 500 °C effectively removes most physisorbed water and a significant fraction of surface –OH species [29]. In the uncalcined Nb<sub>2</sub>O<sub>5</sub>, a weak band in the 1400–1500 cm<sup>-1</sup> region can be ascribed to C=O vibrations from residual oxalate/organic precursor species, in agreement with previous Mo-Nb<sub>2</sub>O<sub>5</sub> reports [16]; this band disappears completely in Nb<sub>2</sub>O<sub>5</sub>-500 and Mo-Nb<sub>2</sub>O<sub>5</sub>-500, confirming almost complete elimination of these residues after heat treatment [30]. Importantly, no new bands attributable

to distinct Mo–O vibrations are observed for Mo-Nb<sub>2</sub>O<sub>5</sub>-500, and the positions of the Nb–O-related bands remain essentially unchanged, supporting that Mo is incorporated into the Nb<sub>2</sub>O<sub>5</sub> lattice without forming separate MoO<sub>x</sub> phases and without disrupting the Nb–O framework.

The surface morphologies of Nb<sub>2</sub>O<sub>5</sub>, Nb<sub>2</sub>O<sub>5</sub>-500, and Mo-Nb<sub>2</sub>O<sub>5</sub>-500 samples were examined using scanning electron microscopy (SEM), as presented in Figure 2. All samples exhibit aggregated nanoparticulate structures, with no distinct long-range ordering, indicating that the hydrothermal synthesis route yields similar granular morphologies regardless of post-treatment or doping. Upon calcination at 500 °C, the Nb<sub>2</sub>O<sub>5</sub>-500 sample shows a slightly more compact and fused particle arrangement compared to the uncalcined Nb<sub>2</sub>O<sub>5</sub>, suggesting enhanced grain connectivity due to thermal sintering. Interestingly, the Mo-Nb<sub>2</sub>O<sub>5</sub>-500 sample displays a surface morphology that remains largely similar to the calcined undoped counterpart but exhibits a visibly rougher texture and more irregular particle boundaries [31, 32].

The microstructure and elemental composition of the Mo-Nb<sub>2</sub>O<sub>5</sub>-500 sample were investigated using field-emission scanning electron microscopy coupled with energy-dispersive X-ray spectroscopy (FESEM-EDS), as presented in Figure 3. The SEM images (Figure 3a) reveal a rough, porous morphology composed of aggregated nanoparticles interspersed with thin, rod-like features [33]. This hierarchical structure likely results from anisotropic crystal growth facilitated by Mo incorporation during hydrothermal synthesis, followed by morphological restructuring during calcination. Such a morphology is advantageous for supercapacitor applications, as it offers a high surface area and promotes efficient electrolyte penetration and ion diffusion. To examine the spatial distribution of the constituent elements, EDS elemental mapping was performed (Figure 3b). The uniform distribution of oxygen (O), niobium (Nb), and molybdenum (Mo) signals confirms the successful and homogeneous doping of Mo into the Nb<sub>2</sub>O<sub>5</sub> framework, with no evidence of phase



**Figure 4.** Cyclic voltammetry (CV) curves of (a)  $\text{Nb}_2\text{O}_5$ , (b)  $\text{Nb}_2\text{O}_5\text{-500}$ , and (c)  $\text{Mo-Nb}_2\text{O}_5\text{-500}$  recorded at various scan rates, demonstrating capacitive behavior and rate responsiveness; (d–f) cycling stability profiles of the respective electrodes evaluated at a scan rate of  $10 \text{ mV s}^{-1}$  over 10 cycles, illustrating electrochemical durability and retention performance following calcination and Mo doping.

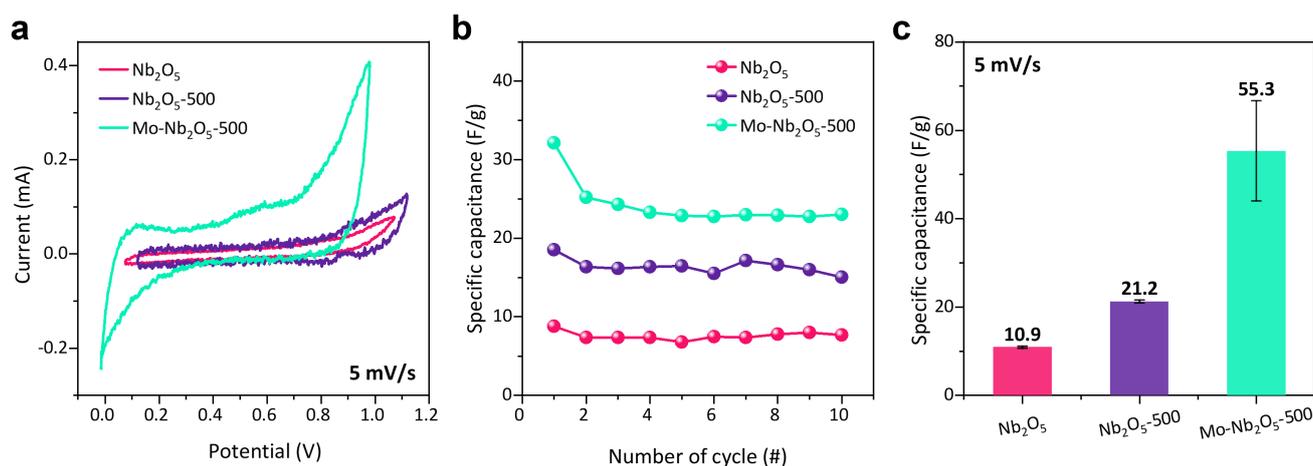
segregation or clustering. This uniformity is critical for ensuring consistent electronic properties throughout the material and maintaining structural integrity during electrochemical cycling.

The corresponding EDS spectra and quantitative analysis (Figures 3c–e) further validate the compositional differences among the samples. Both  $\text{Nb}_2\text{O}_5$  and  $\text{Nb}_2\text{O}_5\text{-500}$  show only Nb and O peaks, with stoichiometries close to the theoretical ratio, although slight shifts in O:Nb ratios are observed after calcination, likely due to structural densification and the removal of surface-bound species. In contrast,  $\text{Mo-Nb}_2\text{O}_5\text{-500}$  exhibits a clear Mo peak, with a measured atomic concentration of 3.84%, confirming successful doping. The concurrent reduction in Nb atomic percentage suggests partial substitution or lattice incorporation of Mo, which can modulate the electronic structure and enhance redox activity.

The cyclic voltammetry (CV) curves of  $\text{Nb}_2\text{O}_5$ ,  $\text{Nb}_2\text{O}_5\text{-500}$ , and  $\text{Mo-Nb}_2\text{O}_5\text{-500}$  measured at various scan rates (5–50 mV/s) are presented in Figures 4a–c. Each sample demonstrates markedly different capacitive behavior. The CV curves of pristine  $\text{Nb}_2\text{O}_5$  (Figure 4a) exhibit a near-rectangular shape with relatively low current response, indicative of electric double-layer capacitance (EDLC)-dominated behavior with limited faradaic contribution, likely due to its poor electronic conductivity and lack of active redox sites [34, 35]. In contrast,

the calcined  $\text{Nb}_2\text{O}_5\text{-500}$  sample (Figure 4b) displays a more distorted CV profile with broader humps, suggesting a transition from purely EDLC to a pseudocapacitive mechanism, likely enabled by enhanced crystallinity and increased surface redox activity due to thermal treatment. The current response is also notably higher than that of uncalcined  $\text{Nb}_2\text{O}_5$ , supporting the positive effect of calcination on electrochemical performance [36].

$\text{Mo-Nb}_2\text{O}_5\text{-500}$  (Figure 4c) exhibits the most pronounced pseudocapacitive behavior, as evidenced by its strong redox peaks and the highest current response among the three. This indicates that molybdenum doping introduces additional redox-active sites and facilitates improved charge storage kinetics, likely due to enhanced electrical conductivity and modified electronic structure [37]. The combination of Mo incorporation and calcination synergistically contributes to the improved capacitive behavior, making  $\text{Mo-Nb}_2\text{O}_5\text{-500}$  a promising candidate for high-performance supercapacitor electrodes. The cycling stability of  $\text{Nb}_2\text{O}_5$ ,  $\text{Nb}_2\text{O}_5\text{-500}$ , and  $\text{Mo-Nb}_2\text{O}_5\text{-500}$  was further assessed over 10 consecutive CV cycles at 10 mV/s, as depicted in Figures 4d, e, and f, respectively. All three samples demonstrate excellent electrochemical reversibility, with minimal distortion and consistent CV profiles across multiple cycles. The near overlap of the CV curves suggests good structural integrity and stable charge-



**Figure 5.** (a) Cyclic voltammetry (CV) comparison of Nb<sub>2</sub>O<sub>5</sub>, Nb<sub>2</sub>O<sub>5</sub>-500, and Mo-Nb<sub>2</sub>O<sub>5</sub>-500 electrodes at a scan rate of 5 mV s<sup>-1</sup>, highlighting the enhanced capacitive response of the doped and calcined sample; (b) specific capacitance evolution over 10 cycles at 10 mV s<sup>-1</sup>, indicating cycling stability; (c) bar chart of specific capacitance at 5 mV s<sup>-1</sup>, showing a pronounced increase for Mo-Nb<sub>2</sub>O<sub>5</sub>-500 relative to the other samples.

discharge behavior, even for the uncalcined Nb<sub>2</sub>O<sub>5</sub> sample. This indicates that the synthesized materials possess inherent robustness under repeated electrochemical operation.

The electrochemical performance of the three electrodes was directly compared through cyclic voltammetry (CV) and specific capacitance evaluation, as shown in Figure 5. The CV curves measured at 5 mV/s (Figure 5a) clearly show differences in capacitive behavior among the samples. Pristine Nb<sub>2</sub>O<sub>5</sub> exhibits the lowest current response and the smallest CV area, indicating poor charge storage performance, which is consistent with its low crystallinity and absence of dopants. The calcined Nb<sub>2</sub>O<sub>5</sub>-500 electrode shows moderate improvement in current density and CV area, reflecting enhanced ion diffusion and conductivity resulting from improved crystallinity after thermal treatment. The Mo-Nb<sub>2</sub>O<sub>5</sub>-500 sample demonstrates a markedly larger CV area and more pronounced redox features, highlighting dominant pseudocapacitive behavior. This enhancement is attributed to the synergistic effect of Mo doping and calcination, which together improve surface reactivity, increase the number of redox-active sites, and enhance electron transport within the Nb<sub>2</sub>O<sub>5</sub> framework.

To evaluate the cycling performance of each electrode, the specific capacitance over 10 consecutive CV cycles at 10 mV/s was monitored (Figure 5b). All samples exhibit stable cycling behavior with minimal capacitance loss, indicating good electrochemical reversibility. Notably, Mo-Nb<sub>2</sub>O<sub>5</sub>-500 retains the highest and most consistent capacitance across cycles, underscoring its superior structural and electrochemical stability [38]. The specific capacitance values calculated from CV curves at 5 mV/s are summarized in Figure 5c. The pristine Nb<sub>2</sub>O<sub>5</sub> electrode delivers a specific capacitance of 10.9 F/g, which increases to 21.2 F/g after calcination. The Mo-doped and calcined sample achieves a significantly enhanced value of 55.3 F/g, confirming the effectiveness of combined doping and thermal treatment in enhancing charge storage capacity. These findings reinforce that Mo-Nb<sub>2</sub>O<sub>5</sub>-500 is the most promising candidate among the tested materials for high-performance supercapacitor electrodes.

#### 4. CONCLUSION

In this study, we successfully synthesized Mo-doped Nb<sub>2</sub>O<sub>5</sub> nanostructures via a hydrothermal method followed by calcination and systematically investigated their structural and electrochemical properties for supercapacitor applications. Scanning electron microscopy revealed that all samples maintained a similar aggregated morphology, with the Mo-doped and calcined sample (Mo-Nb<sub>2</sub>O<sub>5</sub>-500) exhibiting a rougher and more porous surface, which is favorable for ion accessibility. Elemental mapping and EDS analysis confirmed the homogeneous incorporation of Mo into the Nb<sub>2</sub>O<sub>5</sub> matrix without forming secondary phases. X-ray diffraction analysis showed that the pristine Nb<sub>2</sub>O<sub>5</sub> exhibited features of a deformed orthorhombic structure, while the Mo-Nb<sub>2</sub>O<sub>5</sub>-500 sample displayed a pseudohexagonal phase with enhanced crystallinity due to calcination. This structural improvement is associated with better charge transport pathways. Electrochemical measurements demonstrated that both calcination and Mo doping significantly enhanced the capacitive performance of Nb<sub>2</sub>O<sub>5</sub>. The Mo-Nb<sub>2</sub>O<sub>5</sub>-500 electrode achieved a specific capacitance of 55.3 F/g at 5 mV/s, substantially outperforming pristine Nb<sub>2</sub>O<sub>5</sub> (10.9 F/g) and calcined Nb<sub>2</sub>O<sub>5</sub> (21.2 F/g). All samples also exhibited stable cycling behavior, with Mo-Nb<sub>2</sub>O<sub>5</sub>-500 maintaining high capacitance and excellent reversibility over repeated cycles. Overall, the synergistic effect of Mo doping and calcination offers an effective strategy to enhance the electrochemical performance of Nb<sub>2</sub>O<sub>5</sub>-based electrodes, providing valuable insights for the design of advanced pseudocapacitive materials.

#### DATA AVAILABILITY STATEMENT

The datasets generated during and/or analyzed during the current study are available from the corresponding author on reasonable request.

#### CONFLICT OF INTEREST

The authors declare that there are no conflicts of interest associated with this publication, and there has been no significant financial support for this work that could have influenced its outcome.

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